

DFT Studies of the Photocatalytic Properties of BaTiO₃ Doped with V and W for Hydrogen Production

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Abstract. The perovskite BaTiO₃ has been widely used as a catalyst for the synthesis of hydrogen through water splitting. However, the reaction cannot be initiated by visible light due to the large energy band gap of BaTiO₃. This work investigates the prospect of using the most powerful portion of solar energy, visible light, to drive the water splitting reaction. This is achieved by investigating the impact of Mono-doping and Co-doping on the electronic and optical properties of BaTiO₃ using first-principles density functional theory calculations. The doping elements used are Vanadium (V) and Tungsten (W). The computational were performed using the Generalized Gradient Approximation (GGA) and Local Density Approximation (LDA). The results show a reduced band gap and an increase in the optical response.

1 Introduction

The utilisation of solar radiation to perform water-splitting and make hydrogen is widely regarded as a highly advantageous method for generating clean chemical fuels, establishing renewable energy sources, and mitigating the impact of global warming and environmental concerns [1, 2]. Despite significant efforts to create sophisticated materials, the primary obstacle that persists is the reduced efficiency and selectivity for hydrogen (H₂) production using solar energy.

Among these, hydrogen production through semiconductor photocatalysis is one of the most promising pathways for producing green energy [3]. In recent years, researchers have discovered multiple productive methods to enhance the effectiveness of photocatalysis for water splitting. Commonly, precious metals such as platinum (Pt), ruthenium (Ru), rhodium (Rh), copper (Cu), silver (Ag), and gold (Au) are typically incorporated onto the surface of metal oxides to reduce the rate at which electrons and holes recombine [4]. Using the composition of semiconductors is an additional approach for harnessing visible light for hydrogen production through water splitting. When a semiconductor is connected to another semiconductor with a lower conduction band (CB) level, the electrons in the CB of the first semiconductor will go to the CB of the second semiconductor, resulting in effective separation of electrons and holes [5]. In addition, the process of cation and anion doping is highly effective in enhancing hydrogen production when exposed to visible light. This is due to its ability to shift the photo-response of a semiconductor with a broad energy band gap into the visible light spectrum. Experimental evidence has proven that pure BaTiO₃ can operate as a catalyst for the synthesis of hydrogen through water splitting [6, 7]. BaTiO₃ has a reported wide band gap of 3.2 eV, which allows for the absorption of solar light only within the ultraviolet region and no reaction in the visible light region [8]. One way to reduce the band gap

energy and increase the optical response in the visible light region is by doping BaTiO₃ with a foreign element. This study uses such an approach to alter the band gap of BaTiO₃ and increase the optical absorption activity in the visible light region. This is achieved by investigating the impact of Mono-doping on the electronic and optical properties of BaTiO₃ using first-principles density functional theory calculations.

2 Computational details

The geometry optimization has been performed based on the density functional theory using the generalized gradient approximation (GGA) with the Perdew-Burke-Ernzerhof (PBE) functional and norm-conserving pseudopotential implemented in the Cambridge Serial Total Energy Package (CASTEP) code on Material Studio of BIOVIA [9, 10]. The plane wave cutoff energy was set at 380 eV, and the Monkhorst-Pack scheme k-point grid sampling of Brillouin was set at 6x6x6.

3 Results

3.1 Structural Properties

A 2x2x2 supercell of BaTiO₃ was constructed from a unit cell. Figure 1 (a) shows the structure of a pure BaTiO₃ 2x2x2 supercell. Figure 1 (b) shows the structure of a doped BaTiO₃ 2x2x2 supercell system. The doping was done by substituting the Ti atom with either V or W. The lattice parameters of the optimized pure BaTiO₃ structure were calculated. The obtained lattice parameters are listed in Table 1. The results are in consistence with the experimental data as well as the theoretically calculated lattice parameter values [11, 12]. The calculated Ti-O bond length was computed as 2.084 Å and agrees with the experimental value of 2.008 Å.

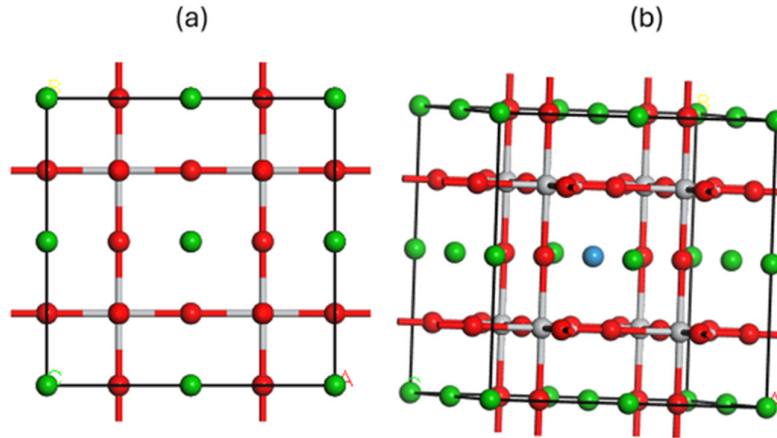


Figure 1: (a) BaTiO₃ 2 × 2 × 2 Supercell structure and (b) Doped BaTiO₃ 2 × 2 × 2 Supercell structure

Table 1: Lattice parameter of pure BaTiO₃

Parameters	This work	Experimental [11]	Theoretical [12]
a (Å)	4.167	3.996	4.030
b (Å)	4.167	3.996	4.030
c (Å)	4.167	3.996	4.030

3.2 Electronic Properties

To understand the electronic properties of BaTiO_3 structure, the band structure calculations were obtained in order to understand the energy levels that the electrons can occupy. Figure 2 (a) shows the band structure calculations of a pure BaTiO_3 $2 \times 2 \times 2$ supercell structure. The dotted black lines at the zero energy represent the Fermi level. The calculation shows an energy difference between the maximum valence band (MVB) and the minimum conduction band (MCB), which is the band gap as 2.219 eV. The band gap is less compared to the reported experimental value of 3.2 eV [8]. The inconsistency in this work and the reported experimental band gap is due to that the GGA calculations underestimate the band gap. However, the underestimation of the band gap does not affect the results of the rest of the calculations. Figure 2 (b) and (c) show the band structures of the V-doped BaTiO_3 $2 \times 2 \times 2$ supercell structure and W-doped BaTiO_3 $2 \times 2 \times 2$ supercell structure, respectively. The calculations show that when BaTiO_3 $2 \times 2 \times 2$ supercell structure is doped with V and W, the band gap reduces to 1.345 and 0.232 eV, respectively.

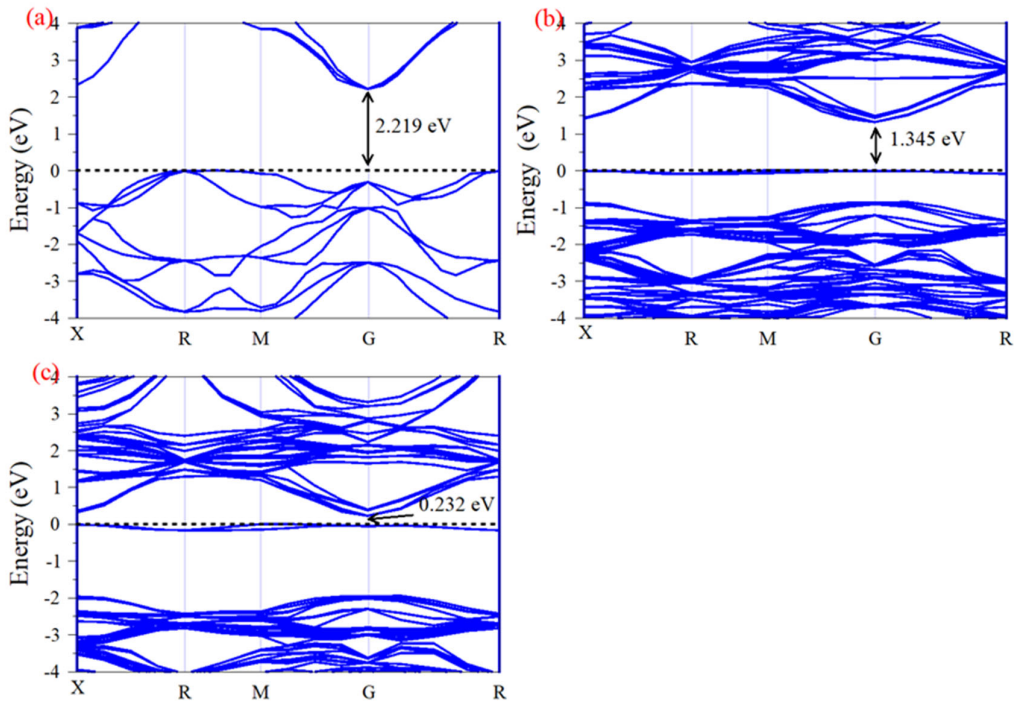


Figure 2: Calculated band structure of (a) BaTiO_3 $2 \times 2 \times 2$ supercell, (b) V- BaTiO_3 $2 \times 2 \times 2$ Supercell, and (c) W- BaTiO_3 $2 \times 2 \times 2$ Supercell.

To study the number of electronic states available within a specific energy range and the contribution of specific orbitals, the total density of states (TDOS) and partial density of states (PDOS) were calculated. Figure 3 (a) depicts the calculated TDOS and PDOS of the BaTiO_3 $2 \times 2 \times 2$ supercell structure. The electron distribution in BaTiO_3 is due to the contribution of the Ba-6s, Ti-3d, and O-2p atomic shells. It is observed that the 3d state of Ti is dominant in the conduction band (CB), while the 2p state of O is dominant in the valence band (VB). The 6s states of Ba have less contribution in the TDOS of BaTiO_3 $2 \times 2 \times 2$ supercell structure. The band gap is observed in the energy range from 0.5 to 2 eV. Figure 3 (b) and (c) depict the calculated TDOS and PDOS of the doped BaTiO_3 $2 \times 2 \times 2$ supercell structure. It is observed in Figures 3 (b) and (c) that the introduction of V and W introduces impurity states at the Fermi level, resulting in a reduction of the band gap.

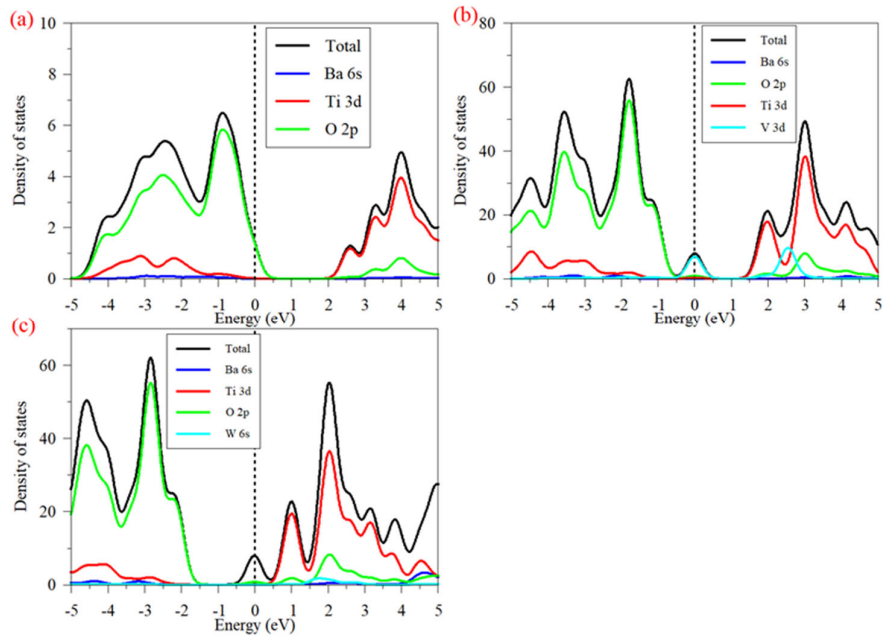


Figure 3: Calculated TDOS and PDOS of (a) BaTiO₃ 2 × 2 × 2 supercell, (b) V- BaTiO₃ 2 × 2 × 2 supercell, and (c) W-BaTiO₃ 2 × 2 × 2 supercell.

3.3 Optical Properties

To investigate how the BaTiO₃ 2x2x2 supercell structure and doped structure respond to electromagnetic radiation, particularly visible, the optical absorptions were calculated. The calculated optical absorptions are shown in Figure 4. Represented by the black curve line, the absorption curve of pure BaTiO₃ 2x2x2 supercell structure shows that it absorbs more in the ultraviolet region and less in the visible region. It is observed that at a wavelength of more than 600 nm of wavelength, pure BaTiO₃ 2x2x2 supercell structure absorption depletes. In comparison, the incorporation of V and W at the Ti sites led to a redshift in the absorption curves.

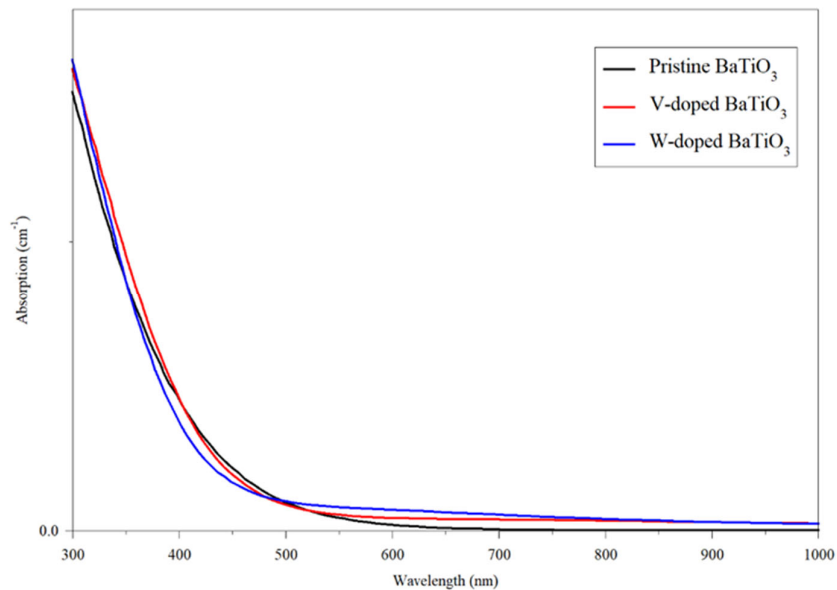


Figure 4: Optical absorption of pure and doped BaTiO₃ 2x2x2 supercell.

4 Conclusion

Using the first principle approach based on density functional theory, the electronic and optical properties of BaTiO₃ were successfully obtained. The calculated lattice parameters were compared with the literature and agreed with it. While GGA underestimated the band gap energy, it did not affect the comparison of the pure structure with the doped structure. The results show that the band gap energy of BaTiO₃ is reduced when BaTiO₃ is doped with V and W. The reduced band gap energy allows electrons to move from VB to CB with less probability of charge recombination. The density of states shows that the reduced band gap energy is due to the impurities that were induced in the Fermi energy. From the optical absorption, the dopants allow BaTiO₃ to have an absorption increase in the visible region. This result enables BaTiO₃ to be effective for water splitting.

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