

# Study on the impact of $\text{Pr}^{3+}$ , $\text{Ce}^{3+}$ , and $\text{Pb}^{2+}$ ions on luminescence properties of $\text{BaB}_8\text{O}_{13}:\text{Gd}^{3+}$ for potential applications in phototherapy

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**Abstract.** A series of  $\text{BaB}_8\text{O}_{13}$  phosphors doped with different concentrations of 2 mol.%  $\text{Gd}^{3+}$  ions and 1.0 mol.% co-doped with  $\text{Pr}^{3+}$ ,  $\text{Pb}^{2+}$  and  $\text{Ce}^{3+}$  ions were synthesized by the combustion synthesis method. The X-ray powder diffraction (XRD) analysis confirmed the pristine crystalline structure and uniformity of the compounds, with an average crystalline size of around 33 – 45 nm. Scanning Electron Microscopy (SEM) was performed to study the surface morphology of the compound, and the bandgap was influenced by the incorporation of 2 mol.%  $\text{Gd}^{3+}$  ions and 1.0 mol.% co-doped with  $\text{Pr}^{3+}$ ,  $\text{Pb}^{2+}$  and  $\text{Ce}^{3+}$  as confirmed from the diffuse reflectance data. The photoluminescence (PL) excitation spectra of  $\text{BaB}_8\text{O}_{13}:\text{Gd}^{3+}$  phosphor showed excitation peaks at 274 nm. The crucial role of gadolinium ( $\text{Gd}^{3+}$ ) ions in barium-based hosts lies in their narrow-band emission spectrum, specifically at 311 – 315 nm, which is in the narrow band ultraviolet B (NB-UVB) range and is attributed to the  ${}^6\text{P}_{7/2}$  to  ${}^8\text{S}_{7/2}$  transition and a prominent emission peak at around 347 nm that could result from oxygen vacancy in the  $\text{BaB}_8\text{O}_{13}$  compound. The effects of co-doping sanitizers to  $\text{Gd}^{3+}$  ions on PL emission at 311 nm were as follows:  $\text{Ce}^{3+} > \text{Pr}^{3+} > \text{Pb}^{2+}$ . Through effective energy transfers from the sensitizers, the broad emission at 387 nm was suppressed while the PL emission of the  $\text{Gd}^{3+}$  ion at 311 nm was increased.  $\text{BaB}_8\text{O}_{13}:2\%\text{Gd}^{3+}$  co-doped with 1 mol%  $\text{Ce}^{3+}$  dominating. Through phototherapy, such material-enhanced NB-UVB emission may be used as a viable treatment option for a variety of skin conditions, including vitiligo, eczema, and psoriasis.

## 1. Introduction

Natural sunlight, also known as heliotherapy, was used in ancient times to treat a variety of skin diseases successfully [1]. Nowadays, artificial light, known as phototherapy, is used effectively to treat a whole array of skin diseases with heightened success. Phototherapy is the interaction of light of specific wavelengths on tissues of the skin to inflict a change in its immune response system with reduced erythema side effects [2]. Light-emitting devices with a phosphor-based coating encapsulated on the inner side of the lamp, and with a wavelength of light emission that aligns with the action spectrum of the skin disease, can be used effectively in this case [3]. Phosphor materials that absorb strongly in the UVC region [200 -280 nm] and for emissions in the narrow-band NB-UVB region [311 – 315 nm] are good for the treatment of skin diseases such as psoriasis as it aligns to the action spectrum of the disease. Minimal literature points towards light emissions in the UVA region [320 - 400 nm], which is useful for a broader treatment of other skin diseases. A typical phosphor material is host dependent and, in this regard, barium octa borate  $\text{BaB}_8\text{O}_{13}$  has been chosen as a host because it has high thermal and chemical stability, it has superior optical properties and has strong absorption in the UVC region [4]. Additionally, an ideal dopant such as gadolinium ion ( $\text{Gd}^{3+}$ ), a rare earth ion whose emission was found to be aligned to a NB-UVB region that is attributed to  ${}^6\text{P}_{7/2}$  to  ${}^8\text{S}_{7/2}$  transition for wavelength excitation of 274 nm. In general, these  $4f-4f$  transitions in  $\text{Gd}^{3+}$  ions are weak, sensitizers such as  $\text{Bi}^{3+}$ ,  $\text{Ce}^{3+}$ ,  $\text{Pr}^{3+}$  and  $\text{Pb}^{2+}$  are

incorporated to enhance its emission in the NB-UVB region by these ions effectively absorbing the excitation energy and transferring it non-radiatively to the  $\text{Gd}^{3+}$  activator ion.

In this work,  $\text{BaB}_8\text{O}_{13}$  is synthesized by a high temperature solid-state reaction synthesis method.  $\text{BaB}_8\text{O}_{13}$  is doped with  $\text{Gd}^{3+}$  and co-doped with  $\text{Pr}^{3+}$ ,  $\text{Ce}^{3+}$  and  $\text{Pb}^{2+}$  ions. The structural, morphological and optical properties, determined by X-ray diffraction (XRD), Scanning electron microscopy (SEM) and UV-visible techniques, respectively. The luminescence properties of these phosphor materials are studies using the photoluminescence (PL) techniques.

## 2. Experimental

### Material and synthesis techniques

Barium octaborate  $\text{BaB}_8\text{O}_{13}$  doped with  $\text{Gd}^{3+}$  ions was synthesized by the combustion synthesis method. In preparation of this compound, barium nitrate ( $\text{Ba}(\text{N}_2\text{O})$ ), boric acid  $\text{H}_3\text{BO}_3$  and urea ( $\text{CH}_4\text{N}_2\text{O}$ ) was used as raw materials. Stoichiometric proportions of the precursor's materials were weighted and mixed in distilled water and ground in a mortar and pestle until a homogeneous solution produced. To ensure better homogeneity of the solution, the mixture was then transferred to a crucible and thereafter transferred to the furnace for annealing at a constant temperature of 800 °C for approximately 4 hours. Once the furnace was cooled to room temperature, the mixture was again ground and was made ready for characterization.

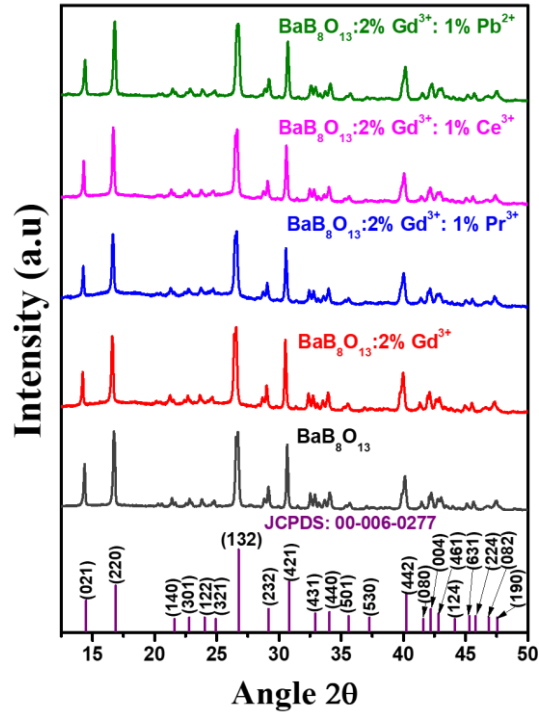
### 3. Material characterization Techniques

X-ray diffraction of the prepared samples was measured using a PANanalytical X'pert powder diffractometer having a wavelength of 1.5406 Å for a  $\text{CuK}_\alpha$  line radiation. SEM measurements were done to determine the surface morphology, and the particle sizes of the prepared materials was obtained using a JSM – 2100 microscope operating at 20 kV voltage. Energy dispersive spectroscopy (EDS) at specific points on the sample were done to obtain the elemental composition of the prepared materials. UV-vis spectroscopy was performed to measure the sample's absorbance or transmittance, done within a spectral range of 200 – 800 nm. Optical measurements were done using a photoluminescent spectrometer, employing a Horiba QN8000 instrument, encased with a xenon lamp as a light source, and a PPD- 850 detector with a fluorescence spectrophotometer with a spectral measuring range of 200 – 600 nm.

## 4. Results and Discussions

### Structural Investigations

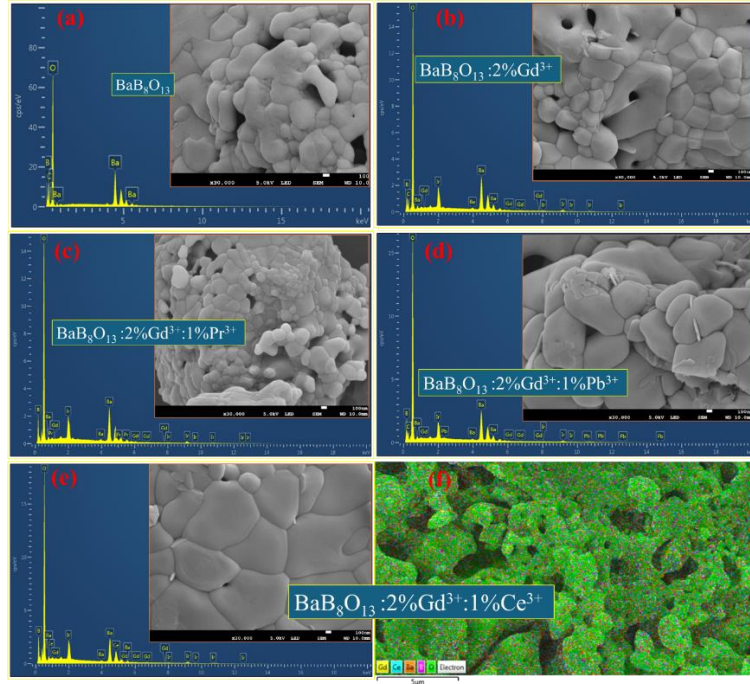
Figure 1 shows the X-ray diffraction patterns of the  $\text{BaB}_8\text{O}_{13}$ ,  $\text{Gd}^{3+}$  (2 mol%) doped  $\text{BaB}_8\text{O}_{13}$ , co-doped with a fixed 1 mol% of  $\text{Ce}^{3+}$ ,  $\text{Pr}^{3+}$ , and  $\text{Pb}^{2+}$  ions, in the range of 10° to 50°. XRD patterns show an orthorhombic crystal structure formation with lattice parameters of  $a = 13.2\text{Å}$ ,  $b = 17.38\text{Å}$  and  $c = 8.56\text{Å}$  [5] having a space group of P4<sub>1</sub>22. These patterns are well compatible with the reference patterns from the JCPDS file #00-006-0277, as indicated in the Figure 1. This pristine orthorhombic structure is seen to have no impurity peaks, confirming a successful incorporation of Gd and co-doped Pr, Ce, and Pb ions. The average crystalline size, determined using the Scherrer equation, was 36.69 nm for the host material. It increased slightly to 37.05 nm with  $\text{Gd}^{3+}$  doping, but decreased to 35.75 nm with  $\text{Pr}^{3+}$ , 35.97 nm with  $\text{Ce}^{3+}$ , and 35.82 nm with co-doping. When a small  $\text{Gd}^{3+}$  ion ( $r = 1.05\text{Å}$ ) substitutes for a larger  $\text{Ba}^{2+}$  ion ( $r = 1.35\text{Å}$ ) in the host lattice, it causes a decrease in the  $\text{Gd}^{3+}-\text{O}^{2-}$  bond distance, which leads to an increase in the crystal field splitting and a shift in the peak positions [3].



**Figure 1.** XRD patterns of  $\text{BaB}_8\text{O}_{13}$ , 2 mol%  $\text{Gd}^{3+}$ -doped  $\text{BaB}_8\text{O}_{13}$ , and co-doped with a fixed 1 mol% of  $\text{Ce}^{3+}$ ,  $\text{Pr}^{3+}$ , and  $\text{Pb}^{2+}$  ions.

### Morphological Investigations

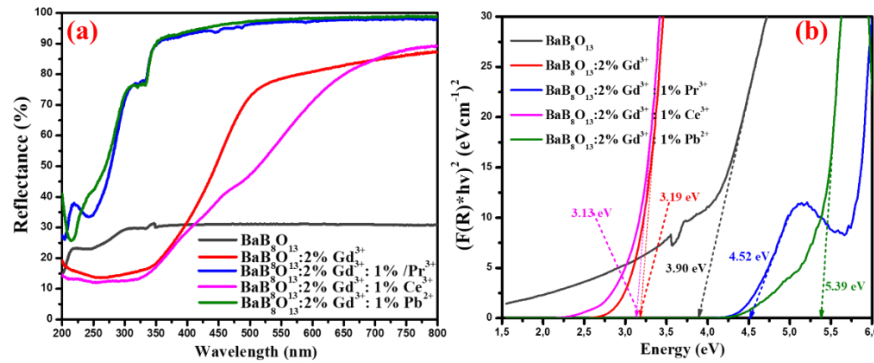
Figure 2 shows the SEM images of the  $\text{BaB}_8\text{O}_{13}$ ,  $\text{BaB}_8\text{O}_{13}:2\%\text{Gd}^{3+}$  and co-doped  $\text{BaB}_8\text{O}_{13}:2\%\text{Gd}^{3+}$  material. SEM images reveal the formation of a pristine crystalline material with uniform crystal structure. Further, it is revealed that the synthesized particles are inhomogeneous, clustered closely and evolve in size and shape when doped with 2 mol%  $\text{Gd}^{3+}$  or co-doped with  $\text{Ce}^{3+}$ ,  $\text{Pr}^{3+}$  or  $\text{Pb}^{2+}$  ions as shown in Figure 2 (a-e). In general, the topology and morphology of the prepared samples change with doping and co-doping. EDS spectra of compounds show the main elements such as Barium, Boron and Oxygen. Carbon and [Iridium](#) (Ir) probably came from the carbon tape mountings and coating layers (used to prevent charging), respectively, in Figure 2 (a-e). Moreover, the selected chemical image mapping image scans describe the uniform distribution of these elements throughout the materials, as depicted in Figure 2 (f).



**Figure 2.** SEM images and respective EDS of the host material  $\text{BaB}_8\text{O}_{13}$  in (a)  $\text{Gd}^{3+}$ - $\text{Gd}^{3+}$ -doped (2 mol% fixed) in (b), 1 mol%  $\text{Pr}^{3+}$  co-doped in (c), 1 mol%  $\text{Pb}^{2+}$  co-doped in (d), and 1 mol%  $\text{Ce}^{3+}$  co-doped in (e) and mapping spectra of 1 mol%  $\text{Ce}^{3+}$  co-doped in (f).

### Optical Investigations

UV-vis diffuse reflectance spectra of the doped and co-doped materials in the wavelength range from 200 to 800 nm were measured, as shown in Figure 3 (a). In the case of the host material, we observe high absorption, making it an ideal candidate for luminescence. Remarkable absorptions are observed for the doped and co-doped samples at their respective absorption edges: around 250 nm for the co-doped samples containing 1 mol%  $\text{Pb}^{2+}$  and 1 mol%  $\text{Pr}^{3+}$  ions, while the redshifted samples doped with 2 mol%  $\text{Gd}^{3+}$  and co-doped with 1 mol%  $\text{Ce}^{3+}$  exhibit absorption edges at approximately 450 nm and 750 nm, respectively. The band gap of these samples is estimated using the Kubelka-Munk absorption coefficient ( $K/S$ ), given by:  $F(R) = K/S = (1-R)^2/2R$ , where  $R$  is the reflectivity,  $K$  is the absorption coefficient and  $S$  is the scattering coefficient [6,7]. The band gap of the samples is obtained by extrapolating the linear section of the Tauc plots, as shown in Figure 3(b). The optical band gap of these samples ranges from 3.13 eV to 5.39 eV.



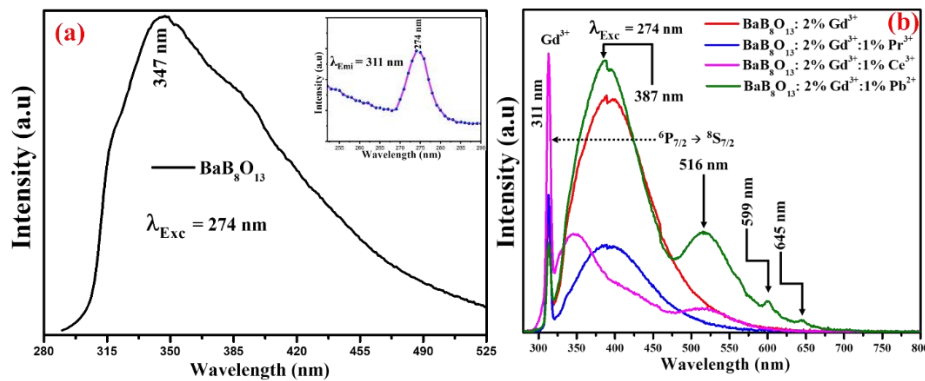
**Figure 3(a):** Reflectance spectra of 2 mol%  $\text{Gd}^{3+}$ -doped  $\text{BaB}_8\text{O}_{13}$  host material, co-doped with a fixed 1 mol% of  $\text{Ce}^{3+}$ ,  $\text{Pr}^{3+}$ , and  $\text{Pb}^{2+}$  ions, while **Figure 3(b)** shows the Tauc plots of the same prepared samples.

The PLE excitation spectra of the  $\text{BaB}_8\text{O}_{13}$  host material, measured at an emission wavelength of 311 nm, is shown in the insert of Figure 4 (a), while its PL spectra is shown in Figure 4(a). The PLE spectra

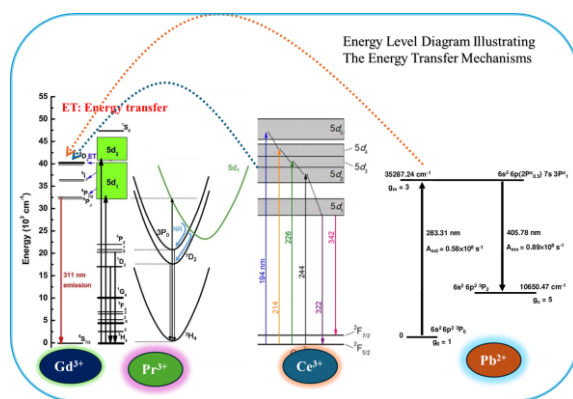
of host shows a broad spectrum in the UVC region with a maximum at 274 nm, attributed to  $^8S_{7/2}$  to  $^6I_J$  transition, corresponding to  $4f$  to  $5d$  orbital transition and a charge transfer from  $O^{2-}$  ( $2p$  orbital) to  $Gd^{3+}$  ( $4f$  orbital) [3]. Figure 4(b) shows the PL emission spectra of  $Gd^{3+}$ -doped  $BaB_8O_{13}$  co-doped with 1 mol% of  $Ce^{3+}$ ,  $Pr^{3+}$ , and  $Pb^{2+}$  ions, recorded under an excitation wavelength of 274 nm. The PL spectrum shows a sharp NB-UVB emission at 311 nm due to  $^8P_{7/2}$  to  $^8S_{7/2}$  transition, followed by a broadband emission centred around 387 nm for all samples except for co-doped 1 mol%  $Ce^{3+}$  which has a maximum around 350 nm. In addition, we observe that the broad emission for the host at 347 nm is red-shifted as the sensitizers are incorporated to around 350-387 nm. It is observed that after doping  $BaB_8O_{13}$  with 2 mol%  $Gd^{3+}$  and co-doping it with 1 mol%  $Ce^{3+}$  and 1% mol  $Pr^{3+}$ , the broadband emissions appear to be suppressed, allowing for the  $Gd^{3+}$  emission at 311 nm to be enhanced, suggesting that whilst the broadband emissions are reduced then the PL emission of  $Gd^{3+}$  is enhanced. It is further noticed that the emissions at 516, 599, and 645 nm are attributed to  $Pb^{2+}$  emission ions is attributed to the transition of the  $Pb^{2+}$  ( $sp-s^2$ ) related charge-transfer state. [8]. Additionally, emissions in the visible regions are also observed for these samples, which makes these samples also applicable for LEDs. The broadband emission is attributed to oxygen vacancies (or oxygen interstitials) and surface defects [9]. when  $Ce^{3+}$  is co-doped with 2 mol%  $Gd^{3+}$ , it has a lower band gap energy which accounts for its enhanced emissions at 311 nm compared to the other co-dopants. In this respect, the 2 mol%  $Gd^{3+}$  doped  $BaB_8O_{13}$  and co-doped with 1 mol%  $Ce^{3+}$  has the strongest emission in the NB-UVB region at 311 nm, while the compound co-doped with  $Pb^{2+}$  and single doped 2%  $Gd^{3+}$  ions has the strongest emission in the broadband region. These compounds, used in tandem as a coating in phosphor lamps, could treat a myriad of skin diseases, instead of psoriasis, as in conventional skin treatments.

### Energy transfer mechanisms

Figure 5 shows the schematic model of the possible energy transfers from the sensitizers ( $Ce^{3+}$ ,  $Pr^{3+}$  and  $Pb^{2+}$ ) ions to the activator  $Gd^{3+}$  ion in the doped  $BaB_8O_{13}$  phosphor material. An observation of the PL spectra in Figure 4, it is observed that enhanced emissions at 311 nm is obtained for the 1 mol%  $Ce^{3+}$  co-doped  $BaB_8O_{13}$ :2% $Gd^{3+}$  phosphor, implying efficient energy transfers occurs mainly from  $Ce^{3+}$  to  $Gd^{3+}$  when equilibrium was established between their excited states. Additionally, some energy transfers of a moderate intensity do occur from  $Pr^{3+}$  to  $Gd^{3+}$  ion at 311 nm. Furthermore, low intensity energy transfer does occur from  $Pb^{2+}$  to  $Gd^{3+}$  ions at 311 nm.



**Figure 4(a):** Shows the PL spectra of the host, while insert shows its PLE spectra and **Figure 4(b)** shows the PL spectra of doped and co-doped samples.



**Figure 5:** Possible schematic model of the energy transfers from the sensitizers ( $\text{Ce}^{3+}$ ,  $\text{Pr}^{3+}$  and  $\text{Pb}^{2+}$ ) to the  $\text{Gd}^{3+}$  activator ions in the  $\text{BaB}_8\text{O}_{13}$  host material [10, 11].

## 5. Conclusion

$\text{Gd}^{3+}$  doped  $\text{BaB}_8\text{O}_{13}$  and co-doped materials with sensitizers ( $\text{Ce}^{3+}$ ,  $\text{Pr}^{3+}$  and  $\text{Pb}^{2+}$ ) show NB-UV-B, UV-A and visible emissions when synthesized by the combustion method at  $800^\circ\text{C}$ . The PL spectra show that the UV-B emissions fall in the 311 nm region when excited by a wavelength of 274 nm. The sharp UV-B emissions at 311 nm corresponds to  $^6\text{P}_{7/2}$  to  $^8\text{S}_{7/2}$  transition from its metastable state to its ground state, and abroad band emission at around 375 nm due to oxygen defect related sources. The XRD data for the  $\text{BaB}_8\text{O}_{13}$  material conforms with JCPDS data having an orthorhombic structure. The SEM results confirm the presence of elements existing in the  $\text{Gd}^{3+}$  doped  $\text{BaB}_8\text{O}_{13}$  materials. UV-vis measurements indicates that these materials are good for luminescence studies due to their strong absorptions in the UV region. The energy transfer between  $\text{Ce}^{3+}$  and  $\text{Gd}^{3+}$  ions makes this compound  $\text{BaB}_8\text{O}_{13}:\text{Ce}^{3+}/\text{Gd}^{3+}$  an ideal candidate for applications in phototherapy lamps, followed by  $\text{BaB}_8\text{O}_{13}:\text{Pr}^{3+}/\text{Gd}^{3+}$  and  $\text{BaB}_8\text{O}_{13}:\text{Pb}^{2+}/\text{Gd}^{3+}$  encapsulated lamps in that order. The broadband emissions of the  $\text{BaB}_8\text{O}_{13}:\text{Pb}^{2+}/\text{Gd}^{3+}$  and  $\text{BaB}_8\text{O}_{13}:\text{Gd}^{3+}$  compounds together with the strong NB-UVB emission of  $\text{BaB}_8\text{O}_{13}:\text{Ce}^{3+}/\text{Gd}^{3+}$  makes them ideal for treating a multitude of skin diseases. Furthermore, emission is also observed in the visible region, thus allowing them to operate in LED lights as well.

## 6. References

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